

## Rare Earth Cobalt Compounds with the $A_2B_{17}$ Structure

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A series of new compounds corresponding to the formula  $R_2Co_{17}$  has been prepared, where R represents the rare earth elements Ce, Pr, Nd, Sm, Gd, Tb, Dy, Ho, Er, Tm and Lu. The compounds  $Pr_2Co_{17}$ ,  $Nd_2Co_{17}$ ,  $Sm_2Co_{17}$ ,  $Gd_2Co_{17}$  and  $Tb_2Co_{17}$  are isomorphous with  $Th_2Zn_{17}$  (space group  $R\bar{3}m$ ), while  $Ho_2Co_{17}$ ,  $Er_2Co_{17}$ ,  $Tm_2Co_{17}$  and  $Lu_2Co_{17}$  were found to have the  $Th_2Ni_{17}$ -type structure (space group  $P6_3/mmc$ ). The patterns of  $Ce_2Co_{17}$  and  $Dy_2Co_{17}$  were explained on the basis of a coexistence of the hexagonal high temperature modification and the rhombohedral low temperature modification. The volume of the formula unit  $R_2Co_{17}$  decreases with increasing atomic number of the rare earth element, except for  $Ce_2Co_{17}$  and  $Pr_2Co_{17}$ . In  $Ce_2Co_{17}$ , cerium appears to occur tetravalent, and in  $Pr_2Co_{17}$ , praseodymium seems to have the effective valency 3.3.

### Introduction

In the course of studies of the magnetic and structural properties of intermetallic compounds between rare earths and cobalt, a series of new compounds of the approximate formula  $R_2Co_{17}$  has been prepared. The results of the magnetic investigation have previously been presented (Strnat, Hoffer, Ostertag & Olson, 1966). It is the purpose of this paper to report on the crystallographic data of this new group of compounds.

The structures of two rare earth cobalt compounds of the composition  $R_2Co_{17}$  have already been reported. Savitskii, Terekhova, Burov, Markova & Naumkin (1962) found  $Ce_2Co_{17}$  to have the  $Th_2Ni_{17}$  structure ( $a=8.36$ ,  $c=8.10$  Å) and Zarechnyuk & Kripyakevich (1963) confirmed these observations. The crystallographic data on  $Gd_2Co_{17}$  show discrepancies; Savitskii, Terekhova & Burov (1962) observed the phase  $Gd_2Co_{17}$  in the course of phase diagram studies and reported it to be isostructural with  $Th_2Zn_{17}$  ( $a=8.55$ ,  $c=12.45$  Å); Kripyakevich, Terekhova, Zarechnyuk & Burov (1963) found  $Gd_2Co_{17}$  to have the  $Th_2Ni_{17}$ -type structure ( $a=8.37$ ,  $c=8.14$  Å).

### Experimental

A first series of samples was prepared by levitation melting of stoichiometric amounts of the rare earth element\* and cobalt\*. The samples were heated for 400 hours at 800°C and cooled to room temperature within 4 hours. To study the homogeneity range of the  $R_2Co_{17}$  structures, a second series of samples was synthesized. Six specimens, containing 87.00, 87.50, 88.24, 88.89, 89.47 and 90.00 at.% Co, were prepared for each rare earth-cobalt system. The specimens underwent a homogenization anneal for 200 hours at 1000°C and were subsequently quenched from various tempera-

tures. The composition of the samples was studied after the heat treatment by chemical analysis. All specimens were examined metallographically. At approximately 900°C, the  $R_2Co_{17}$  structures appear to have their homogeneity region between 87.5 and 89.4 at.% Co. Above 900°C, the homogeneity range becomes broader and may even give rise to a continuous solid solubility region between  $RCo_5$  and  $R_2Co_{17}$ . The crystallographic data presented in this paper were obtained from samples having a composition close to the cobalt-rich end of the  $R_2Co_{17}$  homogeneity range. Most of the samples contained a very slight excess of cobalt which could not be detected by X-ray diffraction. Its presence was indicated, however, by magnetic measurements (Curie point of Co) and by metallographic examination. The X-ray investigation was carried out on powders which were obtained by grinding the intermetallic compounds in an agate mortar under carbon tetrachloride. X-ray diffraction photographs were taken using a 114.6 mm diameter Debye-Scherrer camera along with the standard Norelco equipment. Iron-filtered cobalt radiation ( $\lambda=1.7889$  Å) was employed.

The densities of  $Nd_2Co_{17}$  and  $Er_2Co_{17}$  powder specimens were determined as 8.53 and 9.15 g.cm<sup>-3</sup> respectively, by the pycnometric method with n-octane as immersion fluid.

The structures of all compounds were indexed as hexagonal. Table 1 presents the parameters of the  $R_2Co_{17}$  structures. In order to facilitate comparison, the unit-cell dimensions of  $Pr_2Co_{17}$ ,  $Nd_2Co_{17}$ ,  $Sm_2Co_{17}$ ,  $Gd_2Co_{17}$  and  $Tb_2Co_{17}$  are listed for the triply primitive hexagonal cells although the systematic extinctions of reflections other than  $h-k+1=3n$  or  $-h+k+1=3n$  indicate rhombohedral symmetry for these compounds. The lattice parameters  $a_0$  were obtained from graphical extrapolations of  $hk0$  powder data after Nelson & Riley (1945) while the  $c_0$  parameters were obtained from least-squares plots using the so determined  $a_0$  parameters and  $hkl$  powder data with large  $l$ . Table 2 presents the observed and calculated interplanar spacings of  $Pr_2Co_{17}$ ,

\* The rare earth elements and cobalt were of 99.9% nominal purity.

Table 1. *Crystallographic data for R<sub>2</sub>Co<sub>17</sub> compounds*

	$a_0$	$c_0$	$c/a$	Calculated density	Space group	Formula units per unit cell
Ce <sub>2</sub> Co <sub>17</sub>	8.335 ± 0.002 Å	12.153 ± 0.004 Å	1.458	8.73 g.cm <sup>-3</sup>	<i>R</i> $\bar{3}m$	3
Ce <sub>2</sub> Co <sub>17</sub>	8.335 ± 0.002	8.102 ± 0.004	0.972	8.73	<i>P</i> <sub>6<sub>3</sub></sub> / <i>mmc</i>	2
Pr <sub>2</sub> Co <sub>17</sub>	8.415 ± 0.002	12.170 ± 0.004	1.446	8.56	<i>R</i> $\bar{3}m$	3
Nd <sub>2</sub> Co <sub>17</sub>	8.441 ± 0.002	12.181 ± 0.004	1.443	8.55	<i>R</i> $\bar{3}m$	3
Sm <sub>2</sub> Co <sub>17</sub>	8.402 ± 0.002	12.172 ± 0.004	1.449	8.72	<i>R</i> $\bar{3}m$	3
Gd <sub>2</sub> Co <sub>17</sub>	8.361 ± 0.002	12.159 ± 0.004	1.453	8.91	<i>R</i> $\bar{3}m$	3
Tb <sub>2</sub> Co <sub>17</sub>	8.341 ± 0.002	12.152 ± 0.004	1.457	8.98	<i>R</i> $\bar{3}m$	3
Dy <sub>2</sub> Co <sub>17</sub>	8.335 ± 0.002	12.153 ± 0.004	1.458	9.04	<i>R</i> $\bar{3}m$	3
Dy <sub>2</sub> Co <sub>17</sub>	8.335 ± 0.002	8.102 ± 0.003	0.972	9.04	<i>P</i> <sub>6<sub>3</sub></sub> / <i>mmc</i>	2
Ho <sub>2</sub> Co <sub>17</sub>	8.335 ± 0.002	8.101 ± 0.003	0.973	9.07	<i>P</i> <sub>6<sub>3</sub></sub> / <i>mmc</i>	2
Er <sub>2</sub> Co <sub>17</sub>	8.301 ± 0.002	8.100 ± 0.003	0.976	9.18	<i>P</i> <sub>6<sub>3</sub></sub> / <i>mmc</i>	2
Tm <sub>2</sub> Co <sub>17</sub>	8.285 ± 0.002	8.095 ± 0.003	0.977	9.24	<i>P</i> <sub>6<sub>3</sub></sub> / <i>mmc</i>	2
Lu <sub>2</sub> Co <sub>17</sub>	8.247 ± 0.002	8.093 ± 0.003	0.981	9.41	<i>P</i> <sub>6<sub>3</sub></sub> / <i>mmc</i>	2

Table 2. *Powder diffraction data for Pr<sub>2</sub>Co<sub>17</sub>, Nd<sub>2</sub>Co<sub>17</sub>, Sm<sub>2</sub>Co<sub>17</sub>, Gd<sub>2</sub>Co<sub>17</sub> and Tb<sub>2</sub>Co<sub>17</sub>*

<i>hkl</i>	Pr <sub>2</sub> Co <sub>17</sub>		Nd <sub>2</sub> Co <sub>17</sub>		Sm <sub>2</sub> Co <sub>17</sub>		Gd <sub>2</sub> Co <sub>17</sub>		Tb <sub>2</sub> Co <sub>17</sub>		<i>I</i> <sub>obs</sub>
	<i>d</i> <sub>obs</sub>	<i>d</i> <sub>calc</sub>	<i>d</i> <sub>obs</sub>	<i>d</i> <sub>calc</sub>	<i>d</i> <sub>obs</sub>	<i>d</i> <sub>calc</sub>	<i>d</i> <sub>obs</sub>	<i>d</i> <sub>calc</sub>	<i>d</i> <sub>obs</sub>	<i>d</i> <sub>calc</sub>	
101	6.257	6.262	6.257	6.2685	6.238	6.244	6.219	6.221	6.200	6.209	<i>vvw</i>
102	4.670	4.671	4.680	4.6794	4.670	4.667	4.660	4.656	4.649	4.649	<i>vvw</i>
110	4.210	4.208	4.219	4.2210	n.o.	4.201	n.o.	4.180	n.o.	4.170	<i>vvw</i>
201	n.o.	3.491	n.o.	3.5012	3.475	3.485	n.o.	3.470	3.453	3.462	<i>vvw</i>
202	n.o.	3.126	3.128	3.1342	3.110	3.122	3.106	3.110	3.101	4.104	<i>vvw</i>
113	2.916	2.920	2.924	2.9262	2.916	2.918	2.904	2.910	2.900	2.905	<i>m</i>
104	2.806	2.808	2.810	2.8110	2.803	2.807	2.795	2.803	2.799	2.800	<i>w</i>
121	2.681	2.687	2.695	2.6948	2.681	2.682	2.665	2.670	2.659	2.664	<i>vvw</i>
122	2.506	2.509	2.515	2.5164	2.503	2.506	2.492	2.495	2.486	2.490	<i>vw</i>
300	2.426	2.429	2.437	2.4369	2.425	2.425	2.413	2.413	2.408	2.408	<i>m-s</i>
204	2.322	{ 2.335	2.327	{ 2.3397	2.317	{ 2.334	2.310	{ 2.328	2.319	{ 2.325	<i>m</i>
105		{ 2.309		{ 2.3112		{ 2.308		{ 2.305		{ 2.303	
220	2.102	2.104	2.110	2.1105	2.098	2.100	2.088	2.090	2.085	2.085	<i>vs</i>
303	2.083	2.084	2.089	2.0895	2.081	2.082	2.072	2.074	2.068	2.070	<i>vs</i>
124	2.041	2.042	2.044	2.0463	2.041	2.040					<i>w-m</i>
006		{ 2.028		{ 2.0301		{ 2.028	2.030	{ 2.034	2.028	{ 2.030	<i>m-s</i>
205	2.026	{ 2.024	2.029	{ 2.0272	2.025	{ 2.023	2.017	{ 2.019	2.015	{ 2.016	<i>w</i>
132	1.919	1.918	1.924	1.9238	1.916	1.915	1.907	1.907	1.901	1.902	<i>vvw</i>
223	1.867	1.868	1.871	1.8726	1.866	1.865	1.855	1.858	1.855	1.854	<i>m-s</i>
116	1.826	{ 1.827	1.829	{ 1.8295	1.825	{ 1.827	1.820	{ 1.824	1.818	{ 1.822	<i>m</i>
125		{ 1.824		{ 1.8274		{ 1.823		{ 1.818		{ 1.815	
401	1.799	1.802	1.810	1.8075	1.796	1.799	1.790	1.790	1.789	1.786	<i>vvw</i>
402	1.745	1.745	1.751	1.7506	1.742	1.743	1.734	1.735	1.733	1.731	<i>vw</i>
107	1.692	1.691	1.694	1.6928	1.691	1.691	1.689	1.689	1.689	1.688	<i>vw</i>
134	1.684	1.684	1.687	1.6877	1.680	1.682	1.675	1.676	1.673	1.672	<i>w</i>
232	1.613	1.612	1.619	1.6170	1.611	1.610	1.601	1.602	1.598	1.599	<i>vw</i>
207		{ 1.569	1.569	{ 1.5712		{ 1.569	1.564	{ 1.564	1.563	{ 1.564	<i>vvw</i>
404	1.565	{ 1.563		{ 1.5671	1.565	{ 1.561	1.564	{ 1.555		{ 1.552	<i>vw</i>
306		{ 1.557	1.558	{ 1.5598	1.555	{ 1.556	1.552	{ 1.552	1.550	{ 1.550	<i>w-m</i>
135	1.555	{ 1.555		{ 1.5584		{ 1.553		{ 1.548		{ 1.546	<i>w-m</i>
143	1.482	1.481	1.486	1.4848	1.477	1.478	1.470	1.472	1.468	1.469	<i>w-m</i>
127		{ 1.470		{ 1.4724	1.466	{ 1.470	1.461	{ 1.467	1.468	{ 1.465	<i>w-m</i>
234	1.468	{ 1.465	1.471	{ 1.4691		{ 1.463		{ 1.458	1.455	{ 1.455	
226		{ 1.460	1.463	{ 1.4631	1.457	{ 1.459	1.454	{ 1.455	1.451	{ 1.453	<i>m-s</i>
405	1.460	{ 1.459		{ 1.4620		{ 1.457		{ 1.452		{ 1.449	
330	1.403	1.403	1.406	1.4070	1.401	1.400	1.394	1.393	1.392	1.390	<i>w-m</i>
235	1.379	1.378	1.383	1.3815	1.376	1.376	1.372	1.372	1.370	1.369	<i>w</i>
242	1.344	1.343	1.348	1.3474	1.341	1.341	1.336	1.335	1.333	1.332	<i>vvw</i>
333	1.327	1.326	1.329	1.3294	1.322	1.323	1.318	1.318	1.317	1.315	<i>vs</i>
137		{ 1.318	1.319	{ 1.3205	1.313	{ 1.317	1.309	{ 1.314	1.308	{ 1.312	<i>w-m</i>
504	1.316	{ 1.315		{ 1.3181		{ 1.313		{ 1.307		{ 1.305	
119	1.288	1.287	1.288	1.2888	1.290	1.288	1.286	1.286	1.285	1.285	<i>w-m</i>
152	1.278	1.279	1.282	1.2835	1.278	1.278	1.271	1.272	1.268	1.269	<i>vvw</i>
407	1.256	{ 1.258	1.257	{ 1.2602	1.255	{ 1.257	1.250	{ 1.253	1.248	{ 1.251	<i>w-m</i>
244		{ 1.255		{ 1.2582		{ 1.253		{ 1.248		{ 1.245	
146		{ 1.251	1.254	{ 1.2544	1.249	{ 1.250	1.245	{ 1.256	1.242	{ 1.244	<i>vw</i>
505	1.251	{ 1.250		{ 1.2537		{ 1.249		{ 1.244		{ 1.242	
600	1.215	1.215	1.218	1.2184	1.212	1.212	1.207	1.207	1.204	1.204	<i>vs</i>

Table 3. Powder diffraction data for Ho<sub>2</sub>Co<sub>17</sub>, Er<sub>2</sub>Co<sub>17</sub>, Tm<sub>2</sub>Co<sub>17</sub> and Lu<sub>2</sub>Co<sub>17</sub>

hkl	Ho <sub>2</sub> Co <sub>17</sub>		Er <sub>2</sub> Co <sub>17</sub>		Tm <sub>2</sub> Co <sub>17</sub>		Lu <sub>2</sub> Co <sub>17</sub>		I <sub>obs</sub>
	d <sub>obs</sub>	d <sub>calc</sub>	d <sub>obs</sub>	d <sub>calc</sub>	d <sub>obs</sub>	d <sub>calc</sub>	d <sub>obs</sub>	d <sub>calc</sub>	
101	5.381	5.389	5.367	5.3766	5.367	5.369	5.354	5.355	vw
110	4.152	4.168	4.144	4.1505	n.o.	4.143	4.111	4.124	vvw
102	n.o.	3.532	3.522	3.5285	n.o.	3.525	n.o.	3.521	vvw
201	3.292	3.297	2.382	3.2854	3.272	3.280	3.262	3.267	w
112	2.897	2.904	2.888	2.8986	2.885	2.895	2.885	2.888	m
202	n.o.	2.694	2.685	2.6883	2.681	2.685	n.o.	2.678	vvw
121	2.583	2.571	2.574	2.5760	2.568	2.571	2.559	2.561	w
103	2.526	2.529	2.525	2.5276	2.525	2.526	2.524	2.524	w
300	2.400	2.406	2.395	2.3962	2.392	2.392	2.379	2.381	s
122	n.o.	2.263	2.256	2.2563	2.252	2.253	2.245	2.246	vw
203	2.163	2.162	2.156	2.1587	2.156	2.156	2.152	2.153	m
220	2.083	2.084	2.074	2.0752	2.070	2.071	2.060	2.062	vs
302	2.066	2.069	2.060	2.0623	2.057	2.059	2.052	2.052	vs
004	2.024	2.025	2.026	2.0250	2.022	2.024	2.023	2.023	ms
131	1.943	1.944	1.936	1.9360	1.932	1.932	1.920	1.924	vw
123	1.918	1.919	1.914	1.9152	1.913	1.913	1.909	1.908	m
222	1.852	1.853	1.846	1.8469	1.844	1.844	1.835	1.837	m
114	1.821	1.821	1.820	1.8199	1.817	1.818	n.o.	1.816	vvw
132	1.796	1.795	1.789	1.7888	1.785	1.786	1.775	1.779	vvw
204	n.o.	1.766	1.764	1.7642	n.o.	1.763	1.761	1.760	vvw
401	1.760	1.761	1.755	1.7545	1.752	1.751	n.o.	1.744	vw
124	1.623	1.626	1.624	1.6236	1.621	1.622	n.o.	1.619	vvw
231	1.623	1.622	1.615	1.6160	1.613	1.613	1.605	1.606	vvw
105	n.o.	1.581	1.580	1.5803	n.o.	1.579	n.o.	1.579	vvw
140	n.o.	1.575	1.569	1.5636	1.565	1.566	1.559	1.559	vvw
304	1.549	1.549	1.547	1.5466	1.545	1.545	1.543	1.542	w-m
232	n.o.	1.533	1.527	1.5274	1.525	1.525	n.o.	1.519	vvw
403	1.499	1.500	1.495	1.4960	1.495	1.494	1.489	1.489	vvw
205	1.477	1.478	1.477	1.4769	n.o.	1.476	n.o.	1.474	vvw
142	1.469	1.468	1.463	1.4628	1.461	1.460	1.455	1.454	w-m
224	1.453	1.452	1.450	1.4493	1.448	1.447	1.445	1.444	m-s
134	1.423	1.424	1.421	1.4207	1.418	1.419	1.414	1.415	vvw
501	1.421	1.421	1.416	1.4156	1.413	1.413	n.o.	1.407	vw
233	1.412	1.412	1.408	1.4074	1.404	1.405	1.400	1.400	w-m
125	1.392	1.393	1.392	1.3914	1.391	1.390	1.389	1.388	vw
330	1.389	1.389	1.385	1.3835	1.381	1.381	1.375	1.375	w-m
006	n.o.	1.350	1.351	1.3500	n.o.	1.349	n.o.	1.348	vvw
404	n.o.	1.347	1.343	1.3441	n.o.	1.342	n.o.	1.339	vvw
241	1.345	1.345	1.340	1.3398	1.336	1.337	1.331	1.331	vvw
332	1.314	1.314	1.310	1.3092	1.308	1.307	1.303	1.302	s
242	n.o.	1.293	1.288	1.2880	1.286	1.286	1.282	1.280	vvw
116	1.285	1.284	1.284	1.2837	1.283	1.283	1.282	1.282	w-m
234	1.281	1.282	1.276	1.2787	n.o.	1.277	n.o.	1.273	vvw
151	1.281	1.280	1.276	1.2750	1.274	1.273	n.o.	1.267	vvw
503	1.272	1.273	1.266	1.2690	1.265	1.267	1.261	1.262	vvw
135	1.259	1.259	1.258	1.2573	1.257	1.256	1.252	1.253	vvw
152	n.o.	1.235	1.230	1.2301	n.o.	1.228	n.o.	1.223	vvw
243	1.217	1.218	1.214	1.2135	1.216	1.216	1.207	1.207	w-m
405	1.206	1.206	1.203	1.2033	1.202	1.202	n.o.	1.199	vvw
600	1.203	1.203	1.198	1.1980	1.197	1.196	1.190	1.190	vs

Nd<sub>2</sub>Co<sub>17</sub>, Sm<sub>2</sub>Co<sub>17</sub>, Gd<sub>2</sub>Co<sub>17</sub> and Tb<sub>2</sub>Co<sub>17</sub>, Table 3 those of Ho<sub>2</sub>Co<sub>17</sub>, Er<sub>2</sub>Co<sub>17</sub>, Tm<sub>2</sub>Co<sub>17</sub> and Lu<sub>2</sub>Co<sub>17</sub>.

The intensities of reflections of two representative structures, Nd<sub>2</sub>Co<sub>17</sub> and Er<sub>2</sub>Co<sub>17</sub>, were calculated. It was assumed that Nd<sub>2</sub>Co<sub>17</sub>, representing also the similar structures Pr<sub>2</sub>Co<sub>17</sub>, Sm<sub>2</sub>Co<sub>17</sub>, Gd<sub>2</sub>Co<sub>17</sub> and Tb<sub>2</sub>Co<sub>17</sub>, has the Th<sub>2</sub>Zn<sub>17</sub>-type structure, space group  $R\bar{3}m$  ( $D_{3d}^5$ ) (Makarov & Vinogradov, 1956). Er<sub>2</sub>Co<sub>17</sub>, being isostructural with Ho<sub>2</sub>Co<sub>17</sub>, Tm<sub>2</sub>Co<sub>17</sub> and Lu<sub>2</sub>Co<sub>17</sub>, was assumed to have the Th<sub>2</sub>Ni<sub>17</sub>-type structure, space group  $P6_3/mmc$  ( $D_{6h}^4$ ) (Florio, Baenzinger & Rundle, 1956).

The computation of the relative intensities of reflections was carried out according to the equation

$$I_{hkl} = g \cdot L_p \cdot |F^2| \cdot A(\theta)$$

where  $I_{hkl}$  is the relative intensity of a reflection,  $g$  the multiplicity factor,  $F$  the structure factor and  $A(\theta)$  the absorption factor,  $L_p$  is the combined Lorentz-polarization factor. The wavelength used to compute the  $L_p$  factor was 1.7889 Å. A  $\mu_r$  value of 4.2 was used to calculate the absorption factors for Nd<sub>2</sub>Co<sub>17</sub> as well as for Er<sub>2</sub>Co<sub>17</sub>. In the calculation of the structure factors, the atomic scattering factors of Nd and Er as listed in *International Tables for X-ray Crystallography* (1962) and of Co as given by Thomas & Umeda (1957) were used.

The atomic positions of Nd<sub>2</sub>Co<sub>17</sub>, in terms of  $R\bar{3}m$  and hexagonal axes, are:

Nd<sub>I</sub> in 6(c) ( $z = \frac{1}{2}$ ), Co<sub>I</sub> in 6(c) ( $z = 0.1$ ), Co<sub>II</sub> in 9(d), Co<sub>III</sub> in 18(f) ( $x = \frac{1}{3}$ ), Co<sub>IV</sub> in 18(h) ( $x = \frac{1}{6}$ ,  $z = \frac{1}{6}$ ).

The structure of  $\text{Er}_2\text{Co}_{17}$  has atoms in the following positions of space group  $P6_3/mmc$ :

- $\text{Er}_I$  in  $2(b)$ ,  $\text{Er}_{II}$  in  $2(c)$ ,  $\text{Co}_I$  in  $4(f)$  ( $z=0.89$ ),
- $\text{Co}_{II}$  in  $6(g)$ ,  $\text{Co}_{III}$  in  $12(j)$  ( $x=\frac{1}{3}$ ,  $y=0$ ),
- $\text{Co}_{IV}$  in  $12(k)$  ( $x=\frac{1}{6}$ ,  $z=0$ ).

The results of the computations are shown in Tables 5 and 6. The observed intensities\* are, in general, in

\* The intensities of the diffraction lines were measured with a Joyce Loebel double beam microdensitometer.

Table 4. Powder diffraction data of  $\text{Ce}_2\text{Co}_{17}$  and  $\text{Dy}_2\text{Co}_{17}$

$\text{Ce}_2\text{Co}_{17}$ $d_{\text{obs}}$	$\text{Dy}_2\text{Co}_{17}$ $d_{\text{obs}}$	$I_{\text{obs}}$	$\text{Th}_2\text{Zn}_{17}$ -type $hkl$ $d_{\text{calc}}$		$\text{Th}_2\text{Ni}_{17}$ -type $hkl$ $d_{\text{calc}}$	
5.382	5.386	vw			101	5.389
4.168	4.160	vw	110	4.168	110	4.168
3.458	3.452	vw	201	3.460		
n.o.	3.282	vw			201	3.297
3.101	2.997	vw	202	3.103		
2.900	2.893	m	113	2.904	112	2.904
2.799	2.799	vw	104	2.800		
2.662	2.658	vw	121	2.662		
2.583	2.583	vw			121	2.586
2.528	2.528	vw			103	2.529
2.489	2.489	vw	122	2.489		
2.405	2.402	m-s	300	2.406	300	2.406
2.319	2.317	w-m	{ 204	2.324		
			{ 105	2.303		
2.161	2.160	w-m			203	2.162
2.803	2.081	vs	220	2.084	220	2.084
2.068	2.067	vs	303	2.069	302	2.069
2.027	2.027	m-s	{ 124	2.030	004	2.025
			{ 006	2.025		
2.016	2.016	vw	205	2.016		
1.921	1.922	w			123	1.921
1.852	1.853	s	223	1.852	222	1.852
1.819	1.817	w-m	{ 116	1.822	114	1.821
			{ 125	1.815		
1.785	1.785	vw	401	1.785		
1.761	1.762	vw			401	1.761
1.728	n.o.	vw	402	1.729		
1.687	n.o.	vw	107	1.688		
1.672	1.672	vw	134	1.672		
1.640	1.641	vw	232	1.641		
1.623	1.624	vw			{ 124	1.626
					{ 231	1.622
1.574	1.574	vw	140	1.575	140	1.575
1.549	1.549	m	{ 404	1.551		
			{ 306	1.549	304	1.549
			{ 135	1.545		
1.467	1.467	m	{ 143	1.468	142	1.468
			{ 127	1.465		
			{ 243	1.454		
1.452	1.451	s	{ 226	1.452	224	1.452
			{ 405	1.449		
1.414	1.415	w-m			233	1.414
1.392	1.393	vw			125	1.393
1.388	1.388	m	330	1.389	330	1.389
1.367	1.368	vw	235	1.369		
1.351	n.o.	vw	009	1.350	006	1.350
1.345	1.346	vw			241	1.345
1.314	1.314	vs	{ 333	1.314	332	1.314
			{ 137	1.312		
1.303	1.304	vw	504	1.304		
1.285	1.286	m	119	1.284	116	1.284
1.281	1.280	vw			{ 234	1.282
					{ 151	1.280
					{ 503	1.275
1.267	1.267	vw	152	1.268		
1.259	1.259	vw			135	1.259
1.250	1.249	vw	407	1.251		
1.242	1.242	w	{ 244	1.244		
			{ 505	1.241		
1.220	1.220	w			243	1.220
1.206	1.206	vw			405	1.206
1.203	1.203	vs	600	1.203	600	1.203

satisfactory agreement with the calculated intensities, although in a few cases discrepancies are obvious. These discrepancies are most likely caused by the fact that practically all Co atoms were assumed to occupy ideal positions. It is thought that a refinement of the Co positions might lead to a considerably closer agreement of  $I_{obs}$  and  $I_{calc}$ .

The powder data of  $Ce_2Co_{17}$  and  $Dy_2Co_{17}$  might be interpreted on the basis of a single-phase structure ( $U_2Zn_{17}$ -type, Makarov & Vinogradov, 1956) with the unit-cell dimensions:  $a=8.335$ ,  $c=24.306$  Å. A two-phase explanation, as presented in Tables 1 and 5, appeared, however, more reasonable since it was observed that a variation of temperature and stoichiometry results in a change of the intensities of one set of reflections ( $Th_2Zn_{17}$ -type) relative to the other ( $Th_2Ni_{17}$ -type). Patterns of samples quenched from various temperatures indicated that the  $Th_2Ni_{17}$  structure is the

high-temperature-stable modification. It might be mentioned that by using quenching techniques, the  $Th_2Ni_{17}$ -type structure could also be obtained for  $Gd_2Co_{17}$  and  $Tb_2Co_{17}$  which have, without any doubt, the  $Th_2Zn_{17}$ -type structure as room-temperature-stable form. The patterns of  $Ce_2Co_{17}$  and  $Dy_2Co_{17}$  showed furthermore that the  $Th_2Zn_{17}$ -type structure was associated with the rare earth rich side, the  $Th_2Ni_{17}$ -type structure with the cobalt-rich side of the homogeneity range. Similar observations have been reported by Zalkin, Sands, Bedford & Krikorian (1961) for the dimorphous behavior of  $Ti_2Be_{17}$  and  $Hf_2Be_{17}$ .

### Discussion

The structure investigation of compounds with the approximate composition  $R_2Co_{17}$  showed that with the rare earths Ce, Pr, Nd, Sm, Gd, Tb, and Dy  $Th_2Zn_{17}$ -

Table 5. Calculated and observed intensities of  $Nd_2Co_{17}$

<i>hkl</i>	$I_{obs}$	$I_{calc} \cdot 10^{-6}$	<i>hkl</i>	$I_{obs}$	$I_{calc} \cdot 10^{-6}$	<i>hkl</i>	$I_{obs}$	$I_{calc} \cdot 10^{-6}$
101	n.o.	1.1	223	45	43.3	501	n.o.	0.2
102	3	4.6	116	7	5.1	502	n.o.	0.8
110	1	0.5	125	15	13.7	330	15	21.7
003	n.o.	0.4	401	3	0.2	208	n.o.	0.8
201	n.o.	0.5	402	4	1.1	235	9	8.7
202	3	2.6	107	4	2.8	241	n.o.	0.3
113	30	26.1	134	8	9.9	009	n.o.	0.9
104	9	11.6	231	n.o.	0.4	242	1	1.5
121	4	0.8	232	5	1.9	333	70	73.5
122	5	3.7	140	n.o.	0.5	137	14	4.0
300	49	54.4	207	2	2.5	504		3.5
204	31	8.6	404	4	4.4	151	n.o.	0.3
105		10.1	306	16	10.4	119	15	17.7
220	72	73.7	135		10.4	152	2	1.5
303	77	76.7	108	n.o.	0.8	407	11	2.0
124	15	13.8	143	12	9.0	244		6.8
006	41	46.1	127	15	4.5	146	2	2.5
205	8	8.1	234		7.8	505	4	4.0
131	n.o.	0.5	226	45	40.5	600	85	94.9
132	1	2.5	405		4.7			

Table 6. Calculated and observed intensities of  $Er_2Co_{17}$

<i>hkl</i>	$I_{obs}$	$I_{calc} \cdot 10^{-5}$	<i>hkl</i>	$I_{obs}$	$I_{calc} \cdot 10^{-5}$	<i>hkl</i>	$I_{obs}$	$I_{calc} \cdot 10^{-5}$	<i>hkl</i>	$I_{obs}$	$I_{calc} \cdot 10^{-5}$
101	3	2.4	221	n.o.	0.0	304	10	7.9	305	n.o.	0.0
110	2	1.6	130	n.o.	0.1	141	n.o.	0.0	241	1	1.2
002	n.o.	0.7	104	n.o.	1.0	232	2	1.9	106	n.o.	0.0
111	n.o.	0.0	131	2	1.9	115	n.o.	0.0	332	45	46.8
200	n.o.	0.0	123	14	15.3	403	2	2.5	150	n.o.	0.0
102	1	2.1	222	18	21.3	205	n.o.	0.7	242	15	1.6
201	3	1.1	114	1	1.3	142	15	13.6	116		13.7
112	25	21.7	400	n.o.	0.0	224	30	28.4	234		2.2
120	n.o.	0.1	303	n.o.	0.0	500	n.o.	0.0	225	4	0.0
003	n.o.	0.0	132	1	2.5	134	3	2.4	151		1.1
202	1	1.3	204	1	0.9	501	1	0.3	503	3	2.1
121	4	3.1	401	1	0.4	233	10	9.0	206	n.o.	0.0
103	3	6.4	230	n.o.	0.1	125	4	2.5	135	2	2.3
300	39	38.8	223	n.o.	0.0	330	12	14.6	144	n.o.	1.1
301	n.o.	0.0	402	n.o.	0.5	331	n.o.	0.0	333	n.o.	0.0
113	n.o.	0.0	124	1	3.0	240	n.o.	0.1	152	1	1.5
122	5	3.9	005	n.o.	0.0	143	n.o.	0.0	243	11	8.2
203	18	4.8	231	2	1.4	502	n.o.	0.4	126	n.o.	0.1
220	49	48.9	133	7	10.9	006	1	0.9	405	1	0.6
302	51	50.1	105	2	0.7	404	1	0.6	600	55	59.6
004	32	30.7	140	2	0.9						

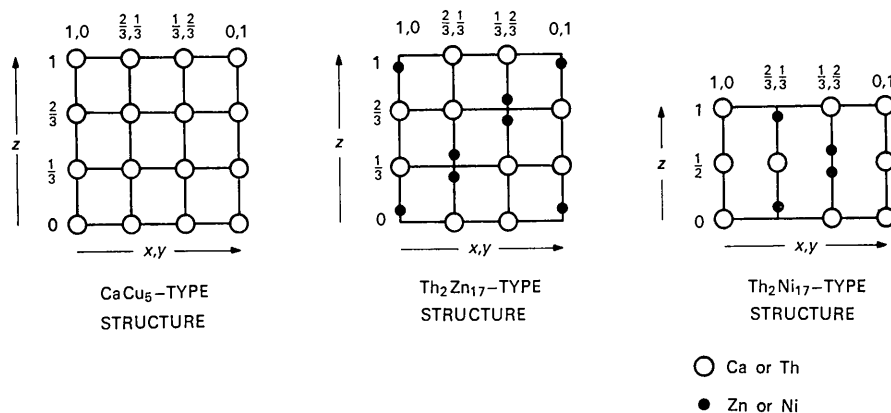


Fig. 1. Relationship between the structure types of  $\text{CaCu}_5$ ,  $\text{Th}_2\text{Zn}_{17}$  and  $\text{Th}_2\text{Ni}_{17}$ .

type structures are formed at room-temperature. The corresponding high-temperature-stable modification is isomorphous with  $\text{Th}_2\text{Ni}_{17}$ . The compounds  $\text{Ho}_2\text{Co}_{17}$ ,  $\text{Er}_2\text{Co}_{17}$ ,  $\text{Tm}_2\text{Co}_{17}$  and  $\text{Lu}_2\text{Co}_{17}$  exhibit the  $\text{Th}_2\text{Ni}_{17}$  structure only. The observed coexistence of the hexagonal and the rhombohedral modification in the specimens of  $\text{Ce}_2\text{Co}_{17}$  and  $\text{Dy}_2\text{Co}_{17}$  is very likely caused by a low transformation temperature which would not allow the crystals to approach complete equilibrium. The transformation temperature may also vary with the stoichiometry which could explain the observed association of the  $\text{Th}_2\text{Zn}_{17}$ -type structure with the rare earth rich side, that of the  $\text{Th}_2\text{Ni}_{17}$ -type structure with the cobalt rich side of the homogeneity range.

Both  $\text{A}_2\text{B}_{17}$ -structures types are closely related. They are derived from an  $\text{AB}_5$ -structure ( $\text{CaCu}_5$ -type) by a systematic replacement of each third A atom by a pair of B atoms within the basal plane. The stacking of the thus formed basic layers in the sequence  $abcabcabc\dots$  leads to the rhombohedral structure type of  $\text{Th}_2\text{Zn}_{17}$ ,

that of  $ababab\dots$  to the hexagonal structure type of  $\text{Th}_2\text{Ni}_{17}$  (Fig. 1).

Fig. 2 shows the variation of the volume per formula unit  $\text{R}_2\text{Co}_{17}$  with the atomic number of the rare earth elements. The volumes decrease from  $\text{Nd}_2\text{Co}_{17}$  to  $\text{Lu}_2\text{Co}_{17}$  as a result of the lanthanide contraction. The cell dimensions of  $\text{Ce}_2\text{Co}_{17}$  and  $\text{Pr}_2\text{Co}_{17}$  are considerably smaller than expected. Wernick & Geller (1959) observed a similar phenomenon for the  $\text{RCO}_5$  compounds (Fig. 2). The unit-cell dimensions of  $\text{Ce}_2\text{Co}_{17}$  and  $\text{Pr}_2\text{Co}_{17}$  suggest that cerium and praseodymium have valencies greater than three. Size considerations and magnetic measurements (Strnat *et al.*, 1966) indicate that Ce has a valency of four in  $\text{Ce}_2\text{Co}_{17}$ , Pr an effective valency of 3.3 in  $\text{Pr}_2\text{Co}_{17}$ .

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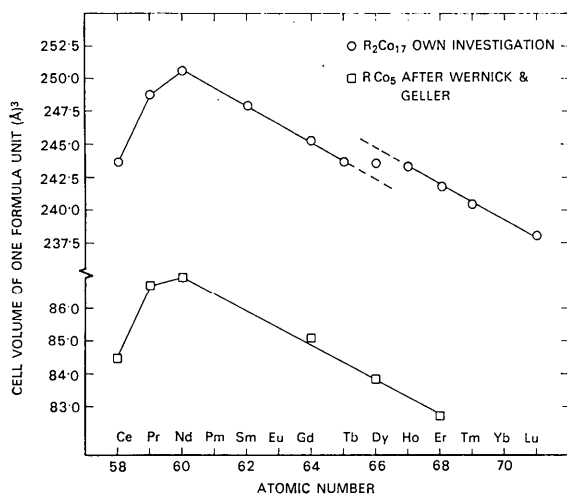


Fig. 2. The variation of the cell volume of one formula unit with the atomic number of the rare earth elements.